Synthesis and growth of materials for solid state lasers: Nd:YLF and Nd:LLW single crystal fibers

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A B S T R A C T

Despite new developments for compact laser devices, the neodymium ion (Nd3+), continues to be the most widely used active laser ion. In this article, the growth of Nd:LiYF 4 (Nd:YLF) and Nd:LiLa(WO4) 2 (Nd:LLW) single crystal fibers growth is reported considering their use as laser media. Homogeneous and good optical quality single crystals of both materials were obtained. Additionally, structural and optical characterizations were performed on Nd:LLW single fibers. The dependence of the atmosphere on Nd:YLF fibers growth were also studied. Dopant distribution was measured along the fibers. Loss and gain tests were performed on the fluoride single crystal fiber to prove that laser action is achievable.

1. Introduction

Since the discovery of the laser, there has been a concerted effort to develop solid-state media. Classical laser materials consist of an active medium, a glass or a bulk crystalline host, to which a dopant is added. Although high perfection bulk crystals are still important for lasers and other applications, in addition to studies to improve their quality, research on single crystal fibers is developing continuously in view of the increasing demand of compact devices [1]. Laser devices benefit by using fiber lasers first in reason of their capability of maintaining high pump intensities over relatively long distances, good beam quality and good thermal management.

Several works on single crystal fiber growth were done for oxide crystals in recent years, improving considerably their optical and structure quality for laser applications. Recent works showed efficient laser action with Nd:YAG [3,4] single crystal fibers. However, fluorides and double tungstates single crystal laser fibers have not often been investigated when compared to other compounds. The Li(RE)(WO4)2 compounds are among the recently studied crystals from the rare earth (Re) double tungstate family materials. Klevtsova and Kozeeva [5] and Klevtsova et al. [6] reported the preparation and structural studies of rare earth and yttrium lithium tungstate crystals. The LiLa(WO4)2 (LLW) crystals have a relatively low melting point (1065 °C), melt congruently and Nd-doped LLW bulk crystals can be grown by the Czochralski (CZ) method. A study of the crystals’ spectroscopic characteristics showed them to be potential solid-state laser crystals [7,8]. To our knowledge there is only one report about the single crystal fiber growth of rare earth double tungstate crystals. Terada et al. [9] reported the preparation of pure and Nd-doped Nd:YLF (Nd:NGW) single fibers by the micro-pulling down (μ-PD) technique, but not Nd:LLW.

The crystal quality of Nd3+-doped LiYF4 (Nd:YLF) has been improved reasonably over the last 40 years, and this laser host of high quality is commercially available, as laser rods or bulk crystals. For optical applications, rare earth doped-YLF is usually grown by the CZ method [10,11]. A small excess of LiF is frequently used in CZ crystal growth to compensate the evaporation of this component during the process and to overcome any variation of composition due to some residual moisture contamination.

Growth by the μ-PD method of pure YLF and Nd and Er-doped YLF single crystal fibers was first reported by our group [12,13]. It was noted that the growth atmosphere plays an important role in the chemical stability and melting behavior of YLF single fibers. Additionally, the growth atmosphere considerably influences the capillarity and wetting effects in YLF single crystal fibers growth. A mixed CF4 and Ar flow was essential for the reduction of oxifluoride formation and for obtaining an appropriate meniscus anchoring for controlled seeding. In these fiber growth experiments, melts with different starting compositions were tested. Moisture contamination and some other complications were avoided by careful cleaning the growth chamber under vacuum (10⁻³ Torr), and replenishing with the inert gas. For YLF single
fibers growth, LiF in excess was needed. The same methodology was applied to pull Nd:YLF single-crystalline fibers with nominal concentrations of 1.7 mol% [14] with similar results.

In this work, we studied the synthesis and growth process of two crystal fibers for use as laser media: Nd:YLF and Nd:LLW. These materials were chosen because of their good laser properties, already demonstrated in bulk crystals and for their low melting point and possibility of doping with several rare earth ions, for studies at different wavelengths.

2. Experimental description

2.1. Fiber growth

YF₃ and NdF₃ were obtained through hydrofluorination of commercial oxides [15]. The LiF compound was pre-purified through a zone melting process [16]. The Nd-doped YLF was prepared by melting the three components under a reactive atmosphere (HF) and used as a starting material for the growth of the fibers. Five mol percent LiF in excess was used at start in the growth charge. The nominal Nd concentrations in two experiments were 0.5 and 1.5 mol%.

The starting materials for the Nd:LLW fiber growth experiments were synthesized through solid state reactions under an air atmosphere. The initial analytical grade chemicals, La₂O₃, Nd₂O₃, WO₃, and Li₂CO₃, were mixed in the appropriate molar ratios to obtain a LiLa0.99Nd0.01(WO₄)₂ compound. The nominal Nd concentration was 1.0 mol%.

The Nd:YLF fiber growth experiments were performed in a modified commercial resistive µ-PD system with atmosphere control. Before the Nd:YLF growth process, the growth chamber was first heated under a vacuum of 10⁻³ Torr and back-filled with ultra-pure argon, following a procedure already described in previous works [12–14]. To improve the purity of the atmosphere, the growth chamber was kept under 10⁻⁷ Torr vacuum, for 24 h. Instead of using a flow of gas during the growth process, the chamber was back-filled with a mixed atmosphere of CF₄ plus Ar and kept at constant pressure (0.1 Kgf/cm²). All fluoride growth experiments were performed under a static atmosphere of this mixture. The Nd:LLW experiments were performed in the same system but under an air atmosphere. Platinum (for YLF) or Platinum/Gold (for LLW) handmade crucibles were used; the fibers were pulled down with rates ranging from 0.06 to 0.2 mm min⁻¹ from a capillary with an inner diameter of 0.8 mm. The visual control of all experiments was performed with a stereo microscope.

2.2. Materials characterization

The fibers were analyzed and photographed through a video-microscope to evaluate macroscopic defects and segregation. X-ray powder diffraction (XRD) analyses of the materials were carried out with a Rigaku RINT 2000/PC diffractometer. The data were collected at room temperature in the range of 15–80° with a step scan of 0.02°/s. All measurements were performed from powders. Micro-X-ray Fluorescence Spectrometry (micro-EDX) was carried out on a Shimadzu spectrometer, model micro-EDX-1300, using a Si(Li) detector operated at 50 kV and 50 μA. Measurements were performed using an incident beam diameter of 50 μm to determine the dopant distribution. Optical absorption spectra were obtained with a Newport spectrometer, model OSM400 VIS-NIR.

Loss and gain tests were performed with a Nd:YLF fiber sample. The two end faces were lightly polished to avoid light scattering; the sample fiber was placed between two plane substrates, and glycercin droplets (index matching fluid) were deposited on the faces, where the fiber touched the substrates. An optical setup was mounted to focus a TEM₀₀ laser beam from a Nd:YLF laser (1053 nm) into the fiber. The laser beam was focused into the fiber through a 30 cm focal length lens, which provides a focus with the Raleigh range equal to half the fiber length and guarantees a collimated beam of approximately constant diameter inside the fiber.

3. Results and discussion

3.1. Nd:LLW fiber growth

Homogeneous 1 mol% doped single-crystalline fibers of Nd:LLW (Fig. 1), of 30 mm length and 0.7–0.8 mm diameters were grown. Transparent and good optical quality fibers (without scattering) were obtained with a pulling rate of 0.06 mm/min. Higher pulling rates induced internal defects resulting in opaque fibers. It is interesting to note that this pulling rate is six times higher than the bulk LLW crystal growth by CZ method, quoted in the literature [7]. Careful control of the power supply was required to keep the fiber diameter constant. The occurrence of liquid overheating increased the meniscus height, and consequently, the fiber diameter decreased. This behavior is typical in materials that do not wet the crucible. An analysis of the XRD data of the grown fibers reveals that fiber crystallization occurs in the scheelite-like tetragonal system, as already reported about the bulk crystals [7].

The non-polarized absorption spectrum of a Nd:LLW single crystal fiber in the wavelength range of 400–1000 nm was measured. Fig. 2 shows the 750–850 nm region in detail. As already reported for CZ crystals, the absorption lines occur at wavelengths near 530, 588, 750, 804 and 872 nm, which are due to the 4fₙ–4fₘ transitions of the Nd³⁺ ion. According to the literature data [7,8], a very weak polarization...
was observed at the 803 nm wavelength, that is, the absorption band at 803 nm is similar for \( \pi \) - and \( \sigma \)-polarizations. Considering the absorption cross section of \( 2.9 \times 10^{-20} \) cm\(^2\) [8], the number density of absorption centers (due to Nd) in the grown fibers was estimated to be \( 0.8 \times 10^{-20} \) cm\(^{-3}\). It is important to mention that such a broad absorption band is suitable for the pumping of an AlGaAs diode-laser that emits an output of 808 nm and is not restricted to the temperature stability of the output wavelength of a diode laser.

3.2. Nd:YLF fiber growth

Nd:YLF single crystalline fibers of \( \sim 80 \) mm length and diameters 0.7–0.8 mm (Fig. 3) were grown using the experimental conditions described above. The YLF phase was directly crystallized on the fiber seeding without the formation of the peritectic transient stage, reported in previous works of our group [12,14]. Such diverse behavior may be attributed to modifications in the methodology of Nd:YLF single fiber growth: the additional high vacuum treatment (10\(^{-7}\) Torr pressure inside the growth chamber) carried out before the fiber pulling and the use of a static atmosphere instead of a continuous gas flow on the growth process.

As already known [17], moisture and oxygen contaminants can degrade the optical quality of Nd:YLF crystals and affect the melting behavior of this compound, from congruent to non-congruent. Thus, improving the atmosphere purity will ensure a higher reduction of spurious moisture contamination, resulting in a uniform and congruent crystallization of the YLF phase on the fiber-seed. The gas flow introduces temperature oscillations, affecting the species transport in the melt and consequently, the peritectic formation with specific complications. Additional studies of flowing, or static atmosphere influence during the seeding process are needed.

The dopant distributions of Y and Nd along the two Nd:YLF fibers were measured by micro-EDX, while Li and F content in YLF was estimated from stoichiometry. The Nd distributions in fibers of 0.5 mol% concentration in the melt (80 mm long) and two samples of 1.5 mol% concentration in the melt (80 and 30 mm long) are presented on Fig. 4.

The Nd distribution of the doped single crystal fiber was quite uniform, with the segregation coefficient \( k \approx 0.33 \), for 0.5 mol% (nominal concentration in the melt). However, at the end of the pulling process (\( \sim 65 \) mm), a small increase in the concentration was noticed, showing the segregation of Nd dopant have \( k < 1 \) in the YLF lattice host [18].

Dopant distribution along the 80 mm fiber (fiber #2 in Fig. 4) had a somewhat larger dispersion and the mean value of \( \sim 0.65 \) mol% Nd concentration in fiber (versus 1.5 mol% in the melt) and the segregation coefficient \( k \approx 0.43 \).

During the unidirectional growth of a crystal, convection in the liquid phase can be driven by several forces. Buoyancy driven convection is very weak in \( \mu \)-PD fiber growth due to the small physical dimensions of the system. However, some influence of the thermocapillary, or the Marangoni convection might have an important role [19]. A theoretical analysis (numerical simulation) would be necessary for better understanding this feature, but this is beyond the scope of the present work. Additional studies on this subject are underway.

3.3. Nd:YLF loss and gain tests

A fiber sample with dimensions of 10 mm in length and 0.7 mm in diameter, with an average Nd concentration of 1.0 mol%, was selected from a single crystal fiber grown with a nominal concentration of 1.5 mol% in the melt.

The loss and gain measurements were performed through the setup shown in Fig. 5 [20]. A fiber loss of 1.90% was obtained, and the loss coefficient was calculated to 0.0192 cm\(^{-1}\). To obtain the gain of the fiber, a 40 W diode laser bar (Coherent) and beam shaper optics were used for pumping. The diode laser was thermally tuned to operate at 805 nm. A Nd:YLF laser beam was injected into the fiber through the 30 cm focal length lens, generating a focus of radius 47 \( \mu \)m. At an average power of 8 mW before the fiber, the calculated intensity was 1.6 kW/cm\(^2\).

The saturated gain \( (g) \) was calculated by the Beer–Lambert law: \( P = P_0 \exp(-gL) \), where \( P \) and \( P_0 \) are the output powers measured at the detector position, with and without pumping, respectively, \( g \) is the gain coefficient and \( L \) is the fiber length. The tested Nd:YLF fiber sample presented a gain coefficient of 0.104 cm\(^{-1}\) (gain minus loss) and a gain per pass of 11%. These values also include the intrinsic losses of the fiber. The calculated non-saturated gain was estimated.
Loss and gain laser tests were performed for fiber samples and it was found that the gain value is much higher than the measured losses. Thus, the laser action of Nd:YLF single crystal grown fibers was proven to be useful for applications.

4. Conclusions

Homogeneous and transparent single crystal fibers of Nd:LLW up to 30 mm long were prepared with appropriate structural and optical quality for laser tests. This media, prepared for the first time in the single crystal fiber shape, may be a good choice for the construction of an efficient and compact laser device, with diode pumping at 808 nm, and emission at 1.06 μm.

In the Nd:YLF single crystal fibers growth experiments, we observed the direct crystallization of the doped-YLF phase on the seeding process, analogous to that which was reported on the CZ growth of this fluoride [18]. This effect mainly related to the atmosphere and a higher melt purity, was obtained with the enhanced vacuum treatment.

Quite uniform Nd dopant distributions were obtained in care-fully elaborated samples, of nominal concentrations 0.5 and 1.5 mol% in the melt. Reproducible dopant distributions of Nd, particularly for melt concentration of 1.5 mol%, have not been achieved. The fiber diameter control in the micro-pulling-down system and the influence of other parameters (pulling rates and the melt convection in the molten zone) has to be further considered. However, the distribution coefficient of Nd, found to be around 0.43 (Fig. 4) was clearly less than one and in fair agreement with some other literature data [18].

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